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## Original Article

# Binary $B_2O_3-Bi_2O_3$ glasses: scrutinization of directly and indirectly ionizing radiations shielding abilities



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## ABSTRACT

For five  $B_2O_3-Bi_2O_3$  glasses, detailedly,  $\gamma$ , neutron, & proton, alpha, and electron (charged particles) shielding efficacies were assessed in this work. The crucial photon attenuating (interaction probabilities) parameter i.e., mass attenuation coefficient ( $\mu/\rho$ ) was calculated by Phy-X/PSD software and procured  $\mu/\rho$  results have been verified through MCNPX, Geant4, & FLUKA codes  $\mu/\rho$  findings from which a quality unanimity among them was noticed over an energy range of 15 KeV–15 MeV. Following  $\mu/\rho$  & linear attenuation coefficient ( $\mu$ ) outcomes,  $Z_{eff}$ ,  $N_{eff}$ , HVL, TVL, and MFP have been reckoned and found that all  $\mu/\rho$ ,  $Z_{eff}$ ,  $N_{eff}$ , HVL, TVL, & MFP highly rest on glass chemical contents & photon energy.  $Z_{eq}$  and by applying geometric progression (G-P) fitting parameters (a, b, c, d, &  $X_k$  coefficients) EBFs & EABFs were appraised at ten specific penetration depths up to 40 mfp, at energy range of 0.015–15 MeV. The inferred RPE quantities confirmed all chosen glasses quintessential absorption competence for lower energy  $\gamma$ -rays. Utilizing SRIM code the mass stopping powers (MSPs) & projected ranges (PRs) for protons ( $(\Psi_p)$  &  $(\Phi_p)$ ) and alpha particles

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( $\Psi_A$ )&( $\Phi_A$ ), and with the help of ESTAR database, electron MSP ( $\Psi_E$ ) & continuous slowing down approximation (CSDA) ranges for electrons have been approximated at 0.015–15 MeV KE. The fast neutron removal cross-sections ( $\Sigma_R$ ) were estimated and obtained  $\Sigma_R$  was diversified at 0.10978–0.12144  $\text{cm}^{-1}$  range relying on  $\text{Bi}_2\text{O}_3$  inclusion in glasses. Based on all acquired outputs, 57.5 $\text{B}_2\text{O}_3$ -42.5 $\text{Bi}_2\text{O}_3$  (mol%) glass possesses superior attenuation capacity for  $\gamma$ -rays and fast neutrons as well as charged particles.

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## 1. Introduction

Despite the increasing demand for renewable energy resources like hydropower, solar energy, wind energy, and geothermal energy utilization, still, coal, oil, gas, and nuclear energy as non-renewable energy resources play a major role in energy supply globally. Further, being a clean energy origin with no pollution, minimal fuel costs, reuse option of recycled fuel, and reliability, nuclear energy for a few decades contributing considerably to the electricity generation. For instance, by the end of 2019, worldwide 450 nuclear power plants are actively functioning producing 10% of the world's electricity [1]. Both  $^{235}\text{U}$  and  $^{239}\text{Pu}$  radioactive sources are widely used nuclear fuels for power creation in nuclear reactors. However, there are some disadvantages associated with nuclear energy, mainly, dangers of accidents, exposure to ionizing radiations ( $\alpha$  &  $\beta$  particles, X-rays,  $\gamma$ -rays, and neutrons), and nuclear waste creation. Besides, different radiopharmaceuticals (e.g.,  $^{99\text{m}}\text{Tc}$ ,  $^{111}\text{In}$ ,  $^{18}\text{F}$ , and  $^{125}\text{I}$ , etc.) that are produced in nuclear reactors are employed in the nuclear medicine field together with imaging techniques to diagnose or treat various diseases [2]. So, for the protection of workers at nuclear power plants & medical staff and patients at where medicinal radiocompounds are used, from the deleterious effects of leaked or scattered and direct radiation exposure that emerge from the radioisotopes the installation of a rightful radiation shielding medium is obligatory. Moreover, to avoid radiation leakage from radioactive waste packages and to safeguard sensitive electronic equipment and astronauts in spacecraft during space missions from high energy radiations requires effective shielding composites [3,4]. Here, the desirable shielding (for  $\gamma$  & neutron) medium must contain both low-Z & high-Z elements in it with fewer irradiation effects on mechanical strengths over time, as alone low-Z compounds show poor photon attenuation capacity.

Classically, concrete (ordinary or heavyweight – holds aggregates like magnetite or barites) is being employed at nuclear power plants and radiotherapy facilities for nuclear radiation shielding as it mainly contains heavy nuclei to absorb  $\gamma$ -rays and  $\text{H}_2\text{O}$  (thus, H atoms) content to attenuate neutrons [5]. Though concrete is widely available, non-toxic, cheap, shows versatility for required any construction design, and needs less maintenance, it has some constraints such as opacity, immovability, cracks formation owing to free and bound  $\text{H}_2\text{O}$  loss by radiation heat, causing radiation leakage over prolonged use, and reduction in mechanical features under high neutron flux [6]. Apart from concretes,

lead (Pb) or Pb found materials (glasses or bricks) are also extensively utilized in the past for shielding purposes at nuclear medical centers & nuclear reactors as Pb ( $Z = 82$  & density ( $\rho$ ) = 11.34  $\text{g}/\text{cm}^3$ ) possesses commendable X-ray and  $\gamma$ -ray attenuation ability. However, unlike concrete, Pb is costly, hard-to-utilize, toxic, and has inferior chemical durability. Particularly, because of Pb's toxicity effects on human health and environment, more recently, it is barely in use as a radiation shield at nuclear medicine centers [7]. From this perspective, numerous research groups have been, in recent years, thoroughly focussed on developing various kinds of Pb-free glasses, for example, Refs [8–11], which could be applied for radiation shielding, as principally glasses have low production costs relying on chemical constituents and are optically transparent. Also, glasses can be produced in large-scale using facile fabrication techniques (i.e., melt-quench or sol-gel approach) with high ' $\rho$ ', good mechanical & thermal strengths, and they are easily recyclable. Vitrally, for a Pb-free glass to be used as a viable radiation shield, it is advisable to choose  $\text{B}_2\text{O}_3$  or  $\text{SiO}_2$ -based glasses rather than  $\text{TeO}_2$  or  $\text{GeO}_2$  containing glasses without much compromising on glass ' $\rho$ ' as the former compounds are much cheaper than the latter ones.

It is widely known that  $\text{B}_2\text{O}_3$  as a classic oxide glass former form glasses at lower melting points than  $\text{SiO}_2$  glasses and are cost-effective than  $\text{GeO}_2$  and  $\text{TeO}_2$  glasses for large scale manufacturing, show fine optical transparency, good mechanical strength, and thermal stability.  $\text{B}_2\text{O}_3$ , as a glass component, considerably drops the  $\text{SiO}_2$  glass melting temperatures, and it is a required component of nuclear waste glasses for high-level radioactive waste immobilization [12]. In 'B' ( $Z = 5$ ) containing  $\text{SiO}_2$  or other oxide glasses & melts, both trigonal  $^3\text{B}$  & tetrahedral  $^4\text{B}$  species of 'B' coevolve, though pure  $\text{B}_2\text{O}_3$  glass contains only  $[\text{BO}_{3/2}]$  structural units [13]. However,  $\text{B}_2\text{O}_3$ -rich glasses exhibit poor chemical durability with quick hydrolysis of  $\text{B}^{[3,4]}-\text{O}-\text{B}^{[3,4]}$  bonds in the presence of  $\text{H}_2\text{O}$  and own relatively larger phonon energy ( $\sim 1300$ – $1500$   $\text{cm}^{-1}$ ) than  $\text{TeO}_2$  or  $\text{GeO}_2$ -based glasses [14]. Here, to improve chemical durability and to reduce phonon energy for practical photonic & optoelectronic applications heavy metal oxides (HMOs) like  $\text{Bi}_2\text{O}_3$  (Bi ( $Z = 83$ )) can be included in an appropriate amount to  $\text{B}_2\text{O}_3$ -based glass compositions wherein  $\text{Bi}_2\text{O}_3$  (Bi–O bond strength  $\sim 81.9$  kcal/mol) acts as both network former & glass modifier, forming  $[\text{BiO}_3]$  pyramidal &  $[\text{BiO}_6]$  octahedral units, accordingly [15].  $\text{Bi}_2\text{O}_3$  also causes a high ' $\rho$ ' & refractive index, enhanced transparency over infrared (IR) region, and excellent nonlinear optical features in the glasses [16].

To utilize distinct substances, say, glasses as  $\gamma$ -ray absorbers, for them, meticulously evaluating the mass attenuation coefficient ( $\mu/\rho$ ) is essential either by proper experimental setup or using appropriate simulation processes (e.g., FLUKA, MNCPIX, Geant4, SRIM, Penelope, MCNP5, PHITS, etc.) and/or suitable theoretical approaches (e.g., XMuDat, XCOM or WinXCOM, MicroShield®, BXCOSM, Phy-X/PSD, etc.). Also, exploring ' $\mu$ ',  $Z_{eff}$ ,  $N_{eff}$ , HVL, TVL, MFP, RPE,  $Z_{eq}$ , EBF & EABF is vital for  $\gamma$ -ray attenuators. Next, estimations of mass stopping powers (MSPs) & projected ranges (PRs) for both protons (p) ( $(\Psi_p)$ & $(\Phi_p)$ ) and alpha ( $\alpha$ ) particles ( $(\Psi_\alpha)$ & $(\Phi_\alpha)$ ), electron ( $e^-$ ) MSP ( $(\Psi_e)$ ) & continuous slowing down approximation (CSDA) ranges for electrons, macroscopic effective removal cross-section for fast neutrons ( $\Sigma_R$ ), and neutron total cross-section ( $\sigma_T$ ) for thermal, slow, and intermediate energy neutrons are desirable to employ them for other directly & indirectly ionizing radiations shielding purpose in addition to  $\gamma$ -rays [17–20]. Concerning  $\gamma$ -rays (photons) as they possess no charge, interact differently than 'p',  $\alpha$  &  $\beta$  particles and there happens three principal interactions (probabilistic, occasional manner) of photons with matter i.e., (a) Photoelectric absorption (PEA) (b) Compton scattering (CS) (incoherent scattering) and (c) pair production (PP) relying on incident photon energy and medium's  $\rho$ ,  $Z_{eff}$  &  $N_{eff}$  whereas p,  $\alpha$  &  $\beta$  particles interact with the medium in a certain, continuous nature. Moreover, Rayleigh scattering (RS) (coherent scattering), Thomson scattering (coherent/elastic scattering), and photonuclear reactions are the other insignificantly considered photon–matter interactions and specifically RS (causes no photon energy loss) phenomenon occur at the lowest  $\gamma$ -ray energies that too can be observed in narrow-beam geometry only.

Considering Pb-free glasses, more recently, Kurtulus & Kavas for  $SiO_2$ – $Al_2O_3$ – $Na_2O$ – $MgO$ – $CaO$ – $SrO$  glass system [21], Olarinoye et al. [22] for  $40TeO_2$ – $(60-x)V_2O_5$ – $xMoO_3$  ( $20 \leq xMoO_3 \leq 60$  mol%) glasses, for  $(25ZnO.75TeO_2)_{100-x}(Ta_2O_5)_x$  ( $x = 0, 1, 2, 3$  mol%) glasses by Kavaz et al. [23], for  $(60-x)B_2O_3+20Na_2O+20BaO+xHgO$  ( $x = 0, 2.5, 5, 7.5, 10, 12.5, 15$  wt %) glasses by Abouhaswa & Kavaz [24], Al-Buriahi et al. [25] for  $(100-x)TeO_2+xZnO+4NiO$  ( $x = 9.6, 19.2, 28.8, 38.4$  mol%) glasses, Akyildirim et al. [26] for  $(75-x)SiO_2-15Na_2O-10CaO-xZrO_2$  ( $x = 0, 1, 3, 5, 7$  mol%) glasses, for  $xBi_2O_3.10Li_2O.0.3CeO_2.9.7MoO_3(80-x)B_2O_3$  ( $x = 0, 10, 15, 30, 45, 60$  wt%) glasses by Kaur et al. [27], Rammah et al. [28] for  $TeO_2$ – $B_2O_3$ – $Bi_2O_3$ – $TiO_2$  glass system, and for  $Bi_2O_3$ – $Li_2O$ – $MnO_2$ – $B_2O_3$  glass system by Kaur et al. [29] relevant nuclear radiation shielding features have been reported.

Being primary purpose of the current study is to examine the suitability of cheaper  $B_2O_3$ – $Bi_2O_3$  glass system as a potential nuclear radiation shield, we scrutinized, employing Phy-X/PSD program within the photon energy range of 15 KeV–15 MeV,  $\mu$ ,  $\mu/\rho$ ,  $Z_{eff}$ ,  $N_{eff}$ , HVL, TVL, MFP, RPE,  $Z_{eq}$ , EBFs, and EABFs – for  $\gamma$ -ray shielding,  $\Psi_p$ ,  $\Phi_p$ ,  $\Psi_\alpha$ ,  $\Phi_\alpha$ ,  $\Psi_e$ , and CSDA ranges – for proton,  $\alpha$ -, and  $e^-$  radiation attenuation by applying SRIM code (for proton &  $\alpha$ -) and ESTAR database (for ' $e^-$ ') within 0.015 KeV–15 MeV kinetic energy (KE) range, and  $\Sigma_R$  – for fast neutrons attenuation using Phy-X/PSD. Likewise,  $\mu/\rho$  quantities are also simulated utilizing MNCPIX (v.2.6.0), Geant4, and FLUKA codes. The estimated lowest HVL & MFP are compared with respective distinct shielding materials

values. Up to 40 mfp PDs (penetration depths), through the G–P fitting formulae, both EBFs and EABFs are reckoned.

## 2. Materials and methods

For chosen all five binary  $B_2O_3$ – $Bi_2O_3$  glasses, the determined ' $\rho$ ' values were acquired from Ref. [30]. The considered five bismuth borate glasses, for simplicity, are indicated as 'B1', 'B2', 'B3', 'B4', and 'B5', accordingly. Related glass chemical compositions (in mol%) and associated computed elemental compositions (in wt%) along with the sample's ' $\rho$ ' selected in this work are given in Table 1. Utilizing  $B_2O_3$  (99.98%) &  $Bi_2O_3$  (99.99%) chemical powders in an accurate ratio, at 1273 K over 24 h heating in capped platinum (Pt) crucibles, after, casting into PtAu95/5 crucibles at 1273 K and then cooling (at 1 K/min rate) to ambient temperature (by cooling the glass melt in a regulated manner), all B1–B5 samples were synthesized [30]. At 290.7 K, through the hydrostatic weighing method, using  $H_2O$  as immersion fluid, all glasses ' $\rho$ ' was ascertained [30]. Following Table 1 data, one can observe that from B1→B5 glass, ' $\rho$ ' raises inevitably because of bigger molecular weight (M.W.) & ' $\rho$ ' of inserted  $Bi_2O_3$  (465.96 g/mol & 8.9 g/cm<sup>3</sup>) instead of lesser M.W. & ' $\rho$ '  $B_2O_3$  (69.63 g/mol & 2.46 g/cm<sup>3</sup>). Here, average M.W. of B1, B2, B3, B4, and B5 glasses are 148.888 g/mol, 168.705 g/mol, 188.522 g/mol, 208.339 g/mol, and 238.0645 g/mol, respectively, and calculated molar volume ( $V_m$ ) is 34.609 cm<sup>3</sup>/mol, 34.620 cm<sup>3</sup>/mol, 35.048 cm<sup>3</sup>/mol, 35.711 cm<sup>3</sup>/mol, 36.961 cm<sup>3</sup>/mol, correspondingly, for the same samples. Thus, while ' $\rho$ ' increases ' $V_m$ ' also enhanced for all B1–B5 glasses. Usually, compared to ' $\rho$ ', ' $V_m$ ' is highly susceptible to structural variations (atoms spatial distribution) among samples as ' $V_m$ ' regulates for various glass chemical constituents' atomic weights.

Statements with related formulae or expressions for  $\mu$ ,  $\mu/\rho$ ,  $Z_{eff}$ ,  $N_{eff}$ , HVL, TVL, MFP, RPE,  $Z_{eq}$ , and geometric progression (G–P) fitting process for EBFs and EABFs prediction were reported earlier in numerous publications [8,9,11,17–20,23,26–28,31], including about  $\Sigma_R$  [9,17,18,20] and they are not rephrased here. When protons and  $\alpha$ -particles or charged particles pass through a shield, by interacting with atoms they gradually lose KE and here, per a unit path length, the KE decrement is expressed as 'linear stopping power (LSP) ( $S(E) = dE/dx$ )' and can be deduced by the following relation [32]:

**Table 1 – Chemical composition (mol%) and elements (wt %) present in the selected binary  $B_2O_3$ – $Bi_2O_3$  glasses, including their density [30].**

Glass code	Glass composition (mol%)		Elemental composition (wt%)			Density (g/cm <sup>3</sup> )
	$B_2O_3$	$Bi_2O_3$	B	Bi	O	
B1	80	20	11.6179	56.1448	32.2373	4.302
B2	75	25	9.6124	61.9371	28.4505	4.873
B3	70	30	8.0285	66.5117	25.4598	5.379
B4	65	35	6.7459	70.2160	23.0381	5.834
B5	57.5	42.5	5.2224	74.6161	20.1615	6.441

$$\frac{dE}{dx} = \frac{4\pi N_A m_e c^2 r_e^2 Z^2}{\beta^2} \rho \frac{Z}{A} \ln 4 \pi \epsilon_0 \frac{\gamma^2 m_e v^3}{Z e^2 f} \quad (1)$$

where  $r_e = \frac{e^2}{4\pi\epsilon_0 m_e c^2}$  = electron radius,  $\beta = \frac{v}{c}$ , and  $N_e = Z \frac{N_A}{A} \rho$ .

Then, mass stopping power (MSP) can be calculated by LSP/ $\rho$ . Further, following Eq. (1), for protons and  $\alpha$ -particles, projected range (PR) in a substance could be estimated by the below expression [28]:

$$R = \int_0^R dx = \int_E^0 \frac{dx}{dE} dE = \int_0^E \frac{dE}{S(E)} \quad (2)$$

where  $S(E) = -\frac{dE}{dx}$ , 'PR' predominantly reckons on the charge, KE, mass, and ' $\rho$ ' of the protons or  $\alpha$ -particles (charged particles).

For electrons (charged particles), it can be approximated that they continuously and progressively degrade in KE during collisions with materials atoms. Here, the term CSDA (continuous slowing down approximation) range represents the integral of the SP reciprocal over KE from a terminal to an initial quantity as below [33]:

$$R_{CSDA} = \int_0^{(E_K)_0} \frac{dE}{S_{total}(E)} \quad (3)$$

where  $R_{CSDA}$  = in a composite, charged particles CSDA diffusion distance,  $E_K$  = initial KE, and  $S_{total}(E)$  = total MSP in line with  $E_K$ .

For all B1–B5 glasses, in this study ( $(\Psi_p)$  &  $(\Phi_p)$ ) and  $(\Psi_A)$  &  $(\Phi_A)$  against KE are computed using SRIM (Stopping and Range of Ions in Matter) codes established by Ziegler et al. [34]. Moreover, the  $\Psi_E$  & CSDA range values versus KE are estimated with the help of the ESTAR database (<https://physics.nist.gov/PhysRefData/Star/Text/ESTAR.html>) [35]. For details about utilized Phy-X/PSD (Photon Shielding and Dosimetry) software readers' can check Ref. [36] and <https://phy-x.net/PSD> web page, as they are not described here. Employed both MCNPX (Monte Carlo N-Particle eXtended) [37] & Geant4 (for GEometry ANd Tracking) [38,39,40] simulation geometry (Figs. 1 and 2) processing specifics are the same as we supplied in our recent works [19,25,31,41] and are not repeated here. Next, Fig. 3 illustrates the simulation setup for exercised FLUKA (FLUktuirende KAskade) code (<http://www.fluka.org/>) and for particulars of this code (created conjunctly by Italian Institute for Nuclear Physics (INFN) & European Organization for Nuclear Research (CERN)) readers' can refer to Refs. [42,43].

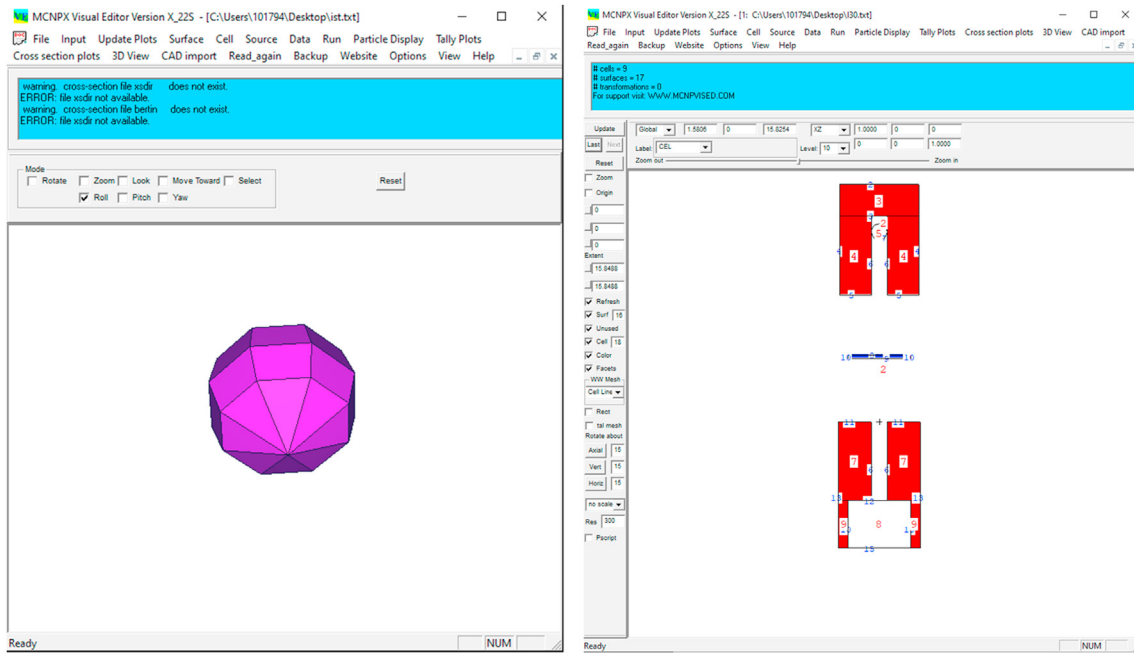
### 3. Results and discussion

In this research, for all B1–B5 samples, the related photon attenuation qualities i.e.,  $\mu$ ,  $\mu/\rho$ ,  $Z_{eff}$ ,  $N_{eff}$ , HVL, TVL, MFP,  $Z_{eq}$ , EBF, EABF, and RPE, unless stated otherwise, are attained within 0.015–15 MeV energy range. Essentially, to consider any medium such as glass or alloy or polymer composite or concrete or ceramic, etc. as a potential  $\gamma$ -ray shield it must possess sufficiently higher ' $\rho$ ', larger  $\mu$ ,  $\mu/\rho$ ,  $Z_{eff}$ ,  $Z_{eq}$ , & RPE, and lower HVL, TVL, MFP, EBFs, & EABFs. Vitaly, at corresponding lower, intermediate, and higher photon energies, certain PEA, CS, and PP phenomenon plays an influential role in any

material. Here, relying on substance 'Z' and incident photon energy (E), PEA  $\propto E^{-3.5}/Z^{4-5}$ , CS  $\propto E^{-1}/Z$ , and PP  $\propto \log E/Z^2$  [8–11,17–29,31]. So, with an increment in E, the odds of PEA reduce, and if the materials' Z enhances, the chances of the PEA increase. Incoherent CS scattering also decreases with E, but not as fast as the PEA. PP mechanism does not happen at  $E < 1.022$  MeV and after this threshold is approached, it occurs probably as E improves.

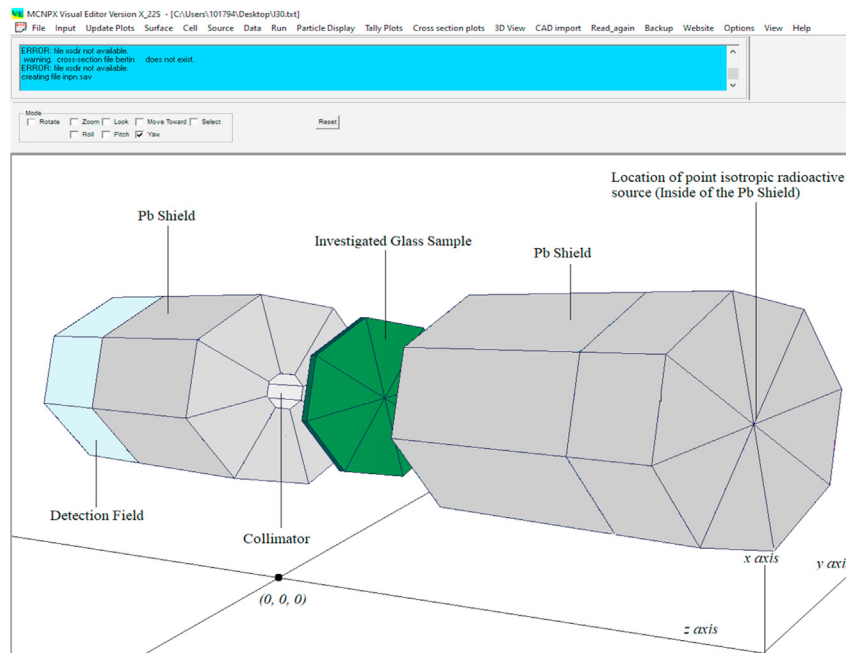
Fig 4 exemplifies the ' $\mu$ ' transitions, calculated using Phy-X/PSD, for all B1–B5 samples. Here, ' $\mu$ ' exhibits an akin drift within examined 'E' range for all chosen glasses and  $Bi_2O_3$  inclusion instead of  $B_2O_3$  causes improvement in ' $\mu$ ' from B1  $\rightarrow$  B5 glass as Bi (Z) = 83 > B (Z) = 5. Consequently, glass B5 (contained 'Bi' wt% = 74.6161 wt%, see Table 1) has the highest ' $\mu$ ' in all investigated samples. 282.87  $cm^{-1}$ , 352.7597  $cm^{-1}$ , 417.5869  $cm^{-1}$ , 477.6746  $cm^{-1}$ , and 559.8516  $cm^{-1}$ , for example, are the derived respective ' $\mu$ ' values for B1, B2, B3, B4, and B5 samples at 15 KeV 'E'. Further, owing to PEA dominance from 15 KeV  $\rightarrow$  0.4 MeV 'E', ' $\mu$ ' reduced sharply, though there occurs a quick rise in it at 0.1 MeV because of Bi element K-absorption edge (=90.53 KeV), for all B1–B5 glasses. In PEA, 'E' can be totally absorbed by an inner shell (e.g., K shell) electron. Next, beyond 0.4 MeV up to 6 MeV 'E' for B1 & B2 samples, and up to 5 MeV 'E' for B3, B4, & B5 glasses, due to CS process influence, ' $\mu$ ' deviated or decreased minimally. CS phenomenon is commonly the very likely kind of interaction in materials. Afterward, above 6 MeV for B1 & B2 glasses and beyond 5 MeV for B3, B4, & B5 samples, up to 15 MeV 'E', on account of PP mechanism, ' $\mu$ ' enhanced slightly. In the PP process, excessive 'E' > 1.022 MeV turns as the KE for positrons & electrons. At 5 & 15 MeV 'E', for instance, for B5 glass, the calculated according ' $\mu$ ' quantities are 0.251877  $cm^{-1}$  & 0.304198  $cm^{-1}$ .

Table 2 (i-iv) presents the pertinent  $\mu/\rho$  outcomes computed by Phy-X/PSD software, MCNPX, Geant4, & FLUKA codes for all B1–B5 glasses and Fig. 5 (a-e) clarifies the achieved  $\mu/\rho$  comparisons for these samples, accordingly. Here, produced by four distinct methods, at all chosen 'E', the acquired  $\mu/\rho$  results are identified to be in tune among themselves. Usually, because of small disparities in the selected geometry and physical models for  $\mu/\rho$  simulations in specific methods, fewer inequalities could possible in the outputs between computational and theoretical techniques. As an example, by Phy-X/PSD, and MCNPX, Geant4, & FLUKA codes at 0.015 MeV 'E', 86.92  $cm^2/g$ , 84.3951  $cm^2/g$ , 85.6117  $cm^2/g$ , & 86.298  $cm^2/g$ , correspondingly, are the deduced  $\mu/\rho$  quantities for sample B5 while these obtained values for the same glass at 15 MeV 'E' with the mentioned same four methods are 0.0472  $cm^2/g$ , 0.0475  $cm^2/g$ , 0.0465  $cm^2/g$ , & 0.0468  $cm^2/g$ , consecutively. It is evident from Fig. 5 & Table 2 that as the 'E' increases, the  $\mu/\rho$  values are reduced for all B1–B5 samples. Moreover,  $Bi_2O_3$  insertion in place of  $B_2O_3$ , i.e., ' $\rho$ ' increment from 4.302  $\rightarrow$  6.441  $g/cm^3$  has led to the  $\mu/\rho$  enhancement from B1  $\rightarrow$  B5 glass, where all samples show an identical  $\mu/\rho$  inclinations with improving 'E'. For instance, utilizing Geant4 code, at 15 KeV 'E', the simulated  $\mu/\rho$  values are 64.8562  $cm^2/g$ , 71.031  $cm^2/g$ , 76.5746  $cm^2/g$ , 80.332  $cm^2/g$ , and 85.6117  $cm^2/g$ , accordingly, for B1, B2, B3, B4, and B5 glasses. At the minimal 'E',  $\mu/\rho$  quantities are high for all B1–B5 samples as the PEA commands this region where within 15 KeV–0.05 MeV 'E' – a



(a)

(b)



(c)

**Fig. 1 – (a) Modeled point isotropic radioactive source obtained from MCNPX Visual Editor (b) 2-D view of simulation setup drawn from MCNPX Visual Editor (version X\_22 S) and (c) 3-D view of employed MCNPX simulation setup (photon transmission setup) obtained from MCNPX Visual Editor.**

quick decline &  $>0.05 \rightarrow 0.3$  MeV 'E' – a small reduction in  $\mu/\rho$  has been noticed. For  $>0.3 \rightarrow (5-8)$  MeV 'E', fewer decrements or fluctuations are observed in  $\mu/\rho$  as the CS phenomenon dominates in this region, and at  $(>5-8) \rightarrow 15$  MeV 'E' range for all studied glasses  $\mu/\rho$  marginally improved due to PP process

dominance in this range. In all  $\mu/\rho$  patterns, the noticed small jump at 100 KeV energy close to the Bi: K-absorption edge can be related to PEA impact. Glass B5 possesses the highest  $\mu/\rho$  for inspected 'E' range among all selected samples, evincing it as an optimum  $\gamma$ -ray shield.

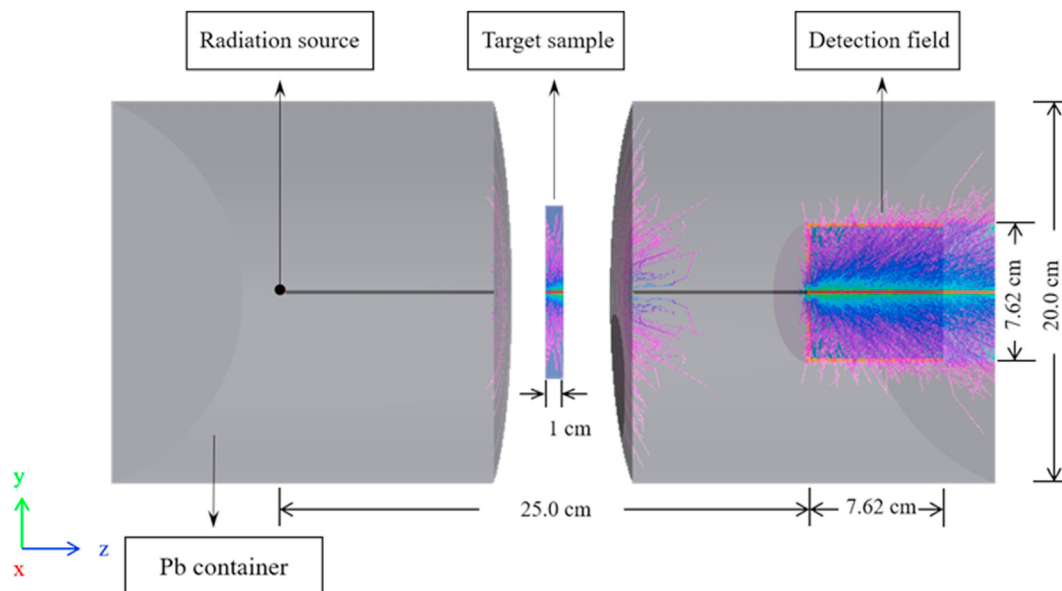


Fig. 2 – Sketch of simulation geometry used for Geant4 code (dimensions are in ‘cm’ units).

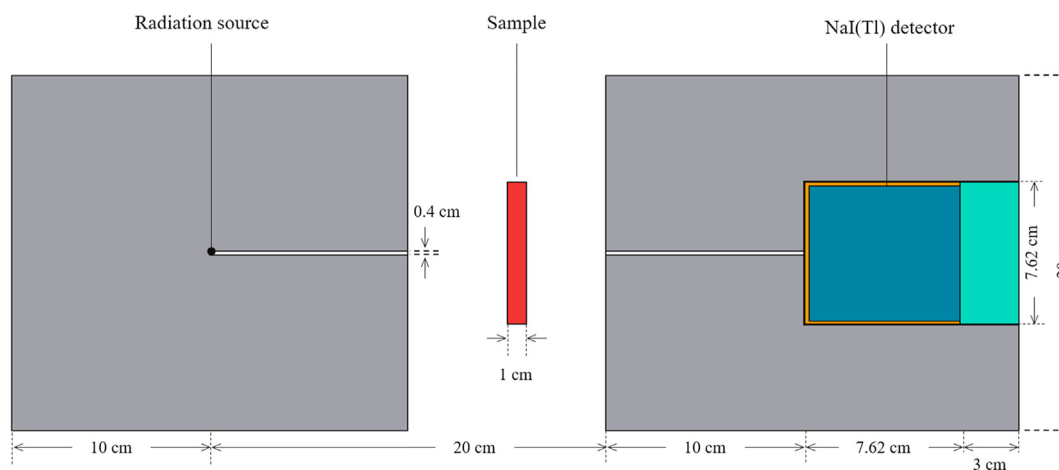


Fig. 3 – Sketch of simulation geometry utilized for FLUKA code (dimensions are in ‘cm’ units).

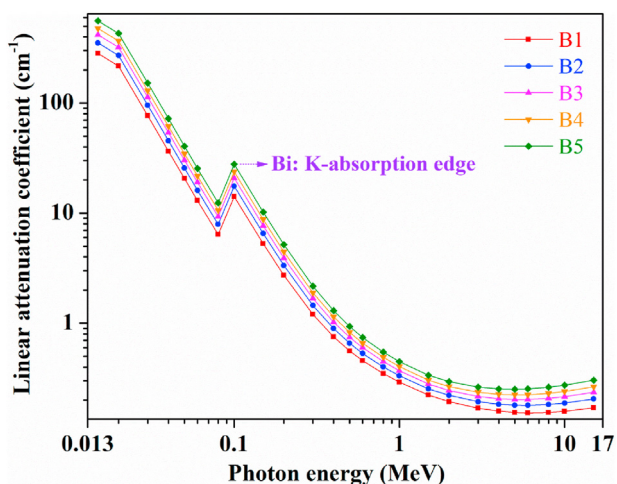


Fig. 4 – Variation of linear attenuation coefficient ( $\mu$ ,  $\text{cm}^{-1}$ ) with photon energy (MeV) for all B1–B5 glasses.

The determined  $Z_{eff}$  &  $N_{eff}$  variations within probed ‘E’ range are displayed in Figs. 6 and 7, correspondingly, for all B1–B5 samples. For any shield, both  $Z_{eff}$  &  $N_{eff}$  values play critical roles in radiation therapy for the estimation of absorbed dose & correct dose delivery required to the patients. Here, for all studied glasses,  $Z_{eff}$  &  $N_{eff}$  profiles reveal alike directions with ‘E’. Following Figs. 6 and 7, one can identify that from B1 → B5 glass, with continuous  $\text{Bi}_2\text{O}_3$  addition in lieu of  $\text{B}_2\text{O}_3$ ,  $Z_{eff}$  increases steadily whereas  $N_{eff}$  values follow an opposite movement to  $Z_{eff}$ , i.e., sample B5 possesses the largest  $Z_{eff}$  and the minimal  $N_{eff}$  quantities in all selected glasses. Generally, the greater the  $Z_{eff}$ , the higher the photons interaction with a medium, leading to less number of  $\gamma$ -rays or X-rays escape from it. For all B1–B5 samples, because of PEA, maximal quantities of  $Z_{eff}$  &  $N_{eff}$  are noticed in the lower ‘E’ range as ‘Bi’ (high-Z) present in all glasses. For instance, 75.61, 77.02, 77.99, 78.7, and 79.47 are the  $Z_{eff}$  quantities at 15 KeV ‘E’ for respective B1, B2, B3, B4, and B5 glasses while at the same ‘E’, for these samples, the calculated relevant  $N_{eff}$  values are

**Table 2 – Mass attenuation coefficients ( $\mu/\rho$ ) of all B1–B5 glasses derived utilizing (i) Phy-X/PSD program (ii) MCNPX (v.2.6.0) (iii) Geant4 and (iv) FLUKA codes.**

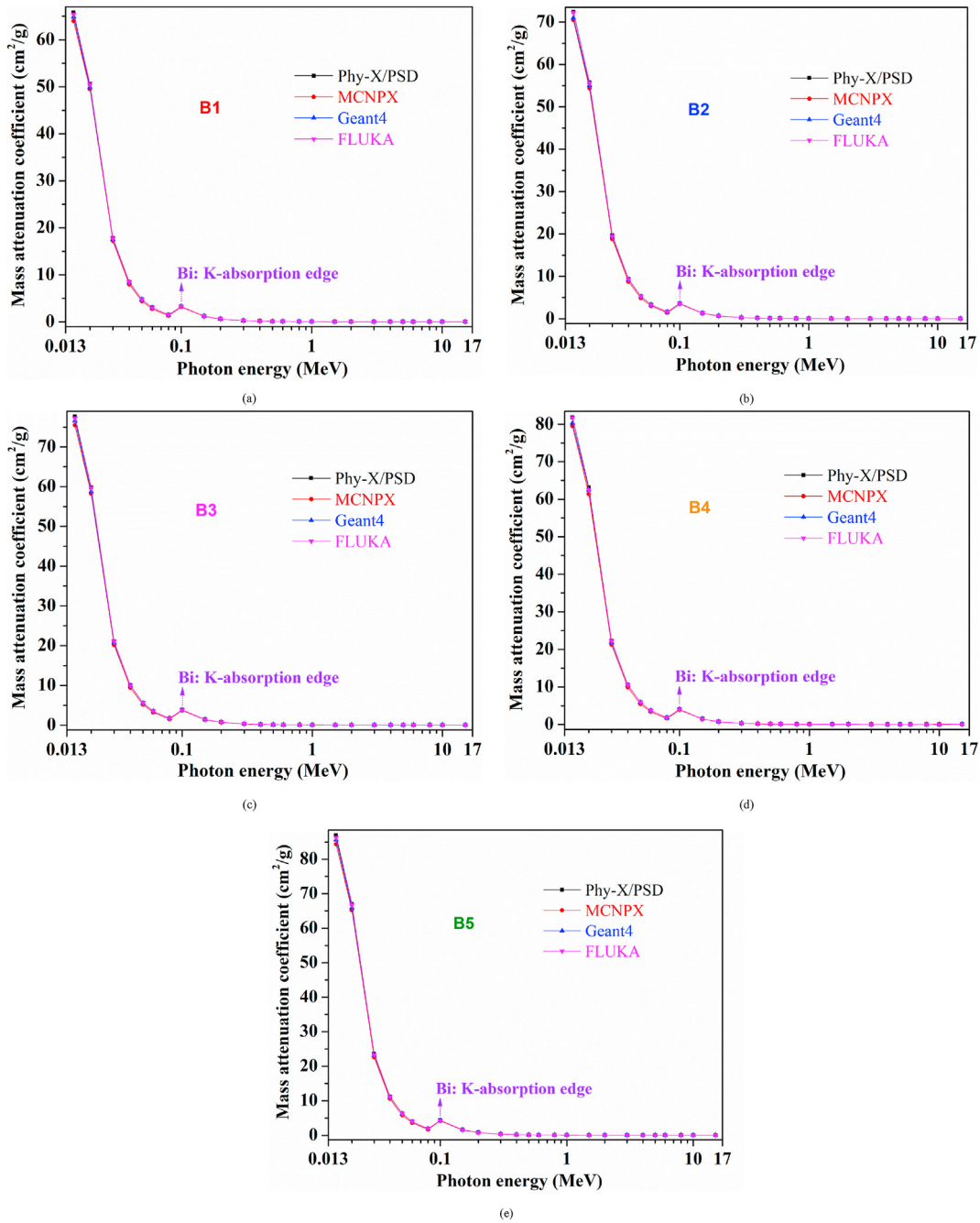
Energy (MeV)	Sample code				
	B1	B2	B3	B4	B5
<b>(i) Phy-X/PSD program</b>					
0.015	65.753	72.391	77.633	81.878	86.92
0.02	50.570	55.716	59.781	63.072	66.981
0.03	17.840	19.647	21.074	22.23	23.603
0.04	8.499	9.3517	10.025	10.57	11.218
0.05	4.792	5.2659	5.6401	5.9432	6.3032
0.06	3.018	3.3106	3.5417	3.7289	3.9513
0.08	1.487	1.6238	1.7318	1.8192	1.9231
0.1	3.287	3.6111	3.8667	4.0737	4.3196
0.15	1.227	1.3404	1.4297	1.5019	1.5877
0.2	0.633	0.6861	0.7279	0.7617	0.8019
0.3	0.280	0.2978	0.3121	0.3237	0.3374
0.4	0.175	0.1838	0.1905	0.1959	0.2023
0.5	0.130	0.1351	0.1388	0.1418	0.1454
0.6	0.106	0.1092	0.1115	0.1133	0.1154
0.8	0.081	0.0824	0.0834	0.0842	0.0851
1	0.068	0.0685	0.0689	0.0693	0.0698
1.5	0.052	0.0521	0.0522	0.0523	0.0524
2	0.045	0.0455	0.0456	0.0458	0.0459
3	0.039	0.0399	0.0403	0.0406	0.0409
4	0.037	0.0378	0.0384	0.0388	0.0394
5	0.036	0.037	0.0378	0.0384	0.0391
6	0.036	0.0369	0.0378	0.0386	0.0394
8	0.036	0.0376	0.0388	0.0397	0.0408
10	0.037	0.0388	0.0402	0.0413	0.0426
15	0.040	0.0422	0.044	0.0455	0.0472
<b>(ii) MCNPX (v.2.6.0) code</b>					
0.015	63.9652	70.4585	75.4524	79.5126	84.3951
0.02	49.5126	54.3622	58.2654	61.3853	65.1956
0.03	17.2413	18.8156	20.1956	21.2426	22.6124
0.04	7.9626	8.7607	9.4120	9.9112	10.5239
0.05	4.4120	4.8512	5.1902	5.4682	5.8004
0.06	2.7398	3.0047	3.2106	3.3841	3.5812
0.08	1.3124	1.4178	1.5198	1.5974	1.6904
0.1	3.1653	3.4853	3.7226	3.9207	4.1571
0.15	1.1672	1.2836	1.3652	1.4284	1.5092
0.2	0.6012	0.6527	0.6884	0.7170	0.7541
0.3	0.2631	0.2791	0.2926	0.3029	0.3150
0.4	0.1662	0.1734	0.1791	0.1841	0.1891
0.5	0.1243	0.1293	0.1316	0.1338	0.1369
0.6	0.1019	0.1051	0.1071	0.1079	0.1101
0.8	0.0799	0.0796	0.0812	0.0810	0.0823
1	0.0674	0.0670	0.0672	0.0682	0.0684
1.5	0.0536	0.0521	0.0523	0.0517	0.0523
2	0.0451	0.0462	0.0462	0.0462	0.0456
3	0.0404	0.0401	0.0411	0.0414	0.0413
4	0.0381	0.0380	0.0391	0.0392	0.0395
5	0.0376	0.0373	0.0382	0.0390	0.0398
6	0.0361	0.0371	0.0386	0.0391	0.0403
8	0.0365	0.0377	0.0391	0.0401	0.0412
10	0.0372	0.0390	0.0412	0.0416	0.0430
15	0.0408	0.0431	0.0443	0.0459	0.0475
<b>(iii) Geant4 code</b>					
0.015	64.8562	71.0310	76.5746	80.3320	85.6117
0.02	49.9339	55.0380	58.7548	62.4240	65.9340
0.03	17.5625	19.3256	20.7288	21.8642	23.1739
0.04	8.3613	9.1913	9.9141	10.4596	10.9824
0.05	4.7503	5.2297	5.5567	5.9093	6.2491
0.06	2.9952	3.2904	3.5007	3.7010	3.8907
0.08	1.4679	1.6071	1.7147	1.8043	1.8896
0.1	3.2549	3.5677	3.8368	4.0517	4.2805
0.15	1.2129	1.3230	1.4161	1.4936	1.5632

**Table 2 (continued)**

0.2	0.6251	0.6761	0.7241	0.7536	0.7916
0.3	0.2759	0.2937	0.3086	0.3199	0.3325
0.4	0.1736	0.1812	0.1882	0.1935	0.2003
0.5	0.1288	0.1336	0.1379	0.1406	0.1442
0.6	0.1054	0.1080	0.1099	0.1123	0.1137
0.8	0.0807	0.0817	0.0823	0.0833	0.0844
1	0.0673	0.0676	0.0682	0.0684	0.0690
1.5	0.0515	0.0516	0.0515	0.0516	0.0519
2	0.0447	0.0451	0.0454	0.0454	0.0456
3	0.0391	0.0396	0.0400	0.0400	0.0404
4	0.0365	0.0373	0.0379	0.0382	0.0389
5	0.0357	0.0365	0.0373	0.0380	0.0386
6	0.0352	0.0365	0.0376	0.0381	0.0388
8	0.0359	0.0371	0.0385	0.0393	0.0402
10	0.0364	0.0383	0.0398	0.0409	0.0423
15	0.0393	0.0417	0.0434	0.0448	0.0465
<b>(iv) FLUKA code</b>					
0.015	65.3893	72.1961	77.1494	81.7293	86.2980
0.02	50.5000	55.6335	59.6379	62.4692	66.7426
0.03	17.7370	19.4593	21.0249	22.1798	23.3200
0.04	8.4384	9.2748	9.9676	10.4932	11.0738
0.05	4.7816	5.2444	5.6355	5.9238	6.2704
0.06	2.9933	3.3047	3.5241	3.7135	3.9320
0.08	1.4868	1.6205	1.7236	1.8183	1.9064
0.1	3.2704	3.5878	3.8373	4.0467	4.2841
0.15	1.2194	1.3323	1.4204	1.4944	1.5789
0.2	0.6312	0.6855	0.7269	0.7589	0.7968
0.3	0.2777	0.2973	0.3111	0.3210	0.3347
0.4	0.1737	0.1827	0.1895	0.1959	0.2018
0.5	0.1297	0.1340	0.1387	0.1415	0.1448
0.6	0.1062	0.1083	0.1110	0.1132	0.1150
0.8	0.0810	0.0821	0.0828	0.0837	0.0845
1	0.0678	0.0680	0.0684	0.0693	0.0695
1.5	0.0519	0.0518	0.0518	0.0523	0.0520
2	0.0451	0.0451	0.0452	0.0456	0.0458
3	0.0391	0.0396	0.0400	0.0402	0.0406
4	0.0369	0.0377	0.0381	0.0386	0.0391
5	0.0360	0.0369	0.0377	0.0383	0.0388
6	0.0355	0.0368	0.0377	0.0385	0.0393
8	0.0359	0.0372	0.0387	0.0397	0.0407
10	0.0368	0.0384	0.0400	0.0413	0.0424
15	0.0397	0.0420	0.0438	0.0453	0.0468

$15.29 \times 10^{23}$ ,  $13.75 \times 10^{23}$ ,  $12.46 \times 10^{23}$ ,  $11.37 \times 10^{23}$ , and  $10.05 \times 10^{23}$  electrons/g. Both  $Z_{eff}$  &  $N_{eff}$  reduces significantly with raising 'E' up to 0.5 MeV with a sudden hike at 0.1 MeV (near Bi: K-absorption edge, by virtue of PEA). Later, owing to the CS process, at  $E > 0.5 \rightarrow 1.5$  MeV, for all chosen glasses, computed both  $Z_{eff}$  &  $N_{eff}$  fluctuates or declines negligibly. The lowest  $Z_{eff}$  &  $N_{eff}$  derived for B1, B2, B3, B4, and B5 samples at 1.5 MeV 'E' are (14.59, 16.49, 18.36, 20.21, and 22.95) & ( $2.951 \times 10^{23}$ ,  $2.943 \times 10^{23}$ ,  $2.932 \times 10^{23}$ ,  $2.921 \times 10^{23}$ , and  $2.903 \times 10^{23}$  electrons/g), accordingly. With further 'E' increment above 1.5 MeV up to 15 MeV,  $Z_{eff}$  &  $N_{eff}$  quantities exhibit a progressive rise due to PP phenomenon predominance in this region. Here, Fig. 6 results firmly endorse that for better photon attenuation potency of  $B_2O_3-Bi_2O_3$  glasses, a greater content of  $Bi_2O_3$  is essential in them. As sample B5 owns relatively bigger  $Z_{eff}$  in all tested glasses, it can be utilized as a superior shield regarding  $\gamma$ -rays. Further,  $N_{eff}$  deviates at  $10.05 \times 10^{23}$  (0.015 MeV 'E') –  $5.237 \times 10^{23}$  (15 MeV 'E') electrons/g range, for glass B5.

Fig. 8 (a), (b) and Fig. 9 illustrates the approximated changes of HVL, TVL, & MFP quantities with 'E', accordingly,



**Fig. 5 – (a–e). Comparison of Phy-X/PSD program, MCNPX, Geant4, and FLUKA codes derived mass attenuation coefficients (cm<sup>2</sup>/g) versus photon energy for all B1–B5 glasses.**

for all B1–B5 samples whereas the corresponding inset plots visualize the enlarged ‘E’ range at 0.029–0.16 MeV. Here, calculated all HVL, TVL, & MFP values exhibit similar deviations within investigated ‘E’ range. From Figs. 8 and 9, one can notice that from 0.015 → 0.1 MeV ‘E’, related HVL, TVL, & MFP are minimal (i.e., a tiny part of ‘cm’), indicating that a smaller thickness B1–B5 glasses are adequate to attenuate photons within this energy range. Then, with increasing ‘E’ beyond 100 KeV up to 6 MeV for samples B1 & B2, and up to 5 MeV for B3, B4, & B5 glasses all HVL, TVL, & MFP increase swiftly attaining the respective maximal values, specifying greater thickness obligation for samples to decrease incident

photons energy. For example, to reduce incoming 5 MeV ‘E’  $\gamma$ -ray to 50% of its initial intensity about 2.752 cm thickness sample B5 is essential. Likewise, the derived MFP quantities at 6 MeV ‘E’ for B1 & B2 glasses and at 5 MeV ‘E’ for B3, B4, and B5 samples are 6.501 cm, 5.56 cm, 4.922 cm, 4.466 cm, and 3.97 cm, accordingly. Thereafter, at >6 MeV ‘E’ (for B1 & B2 samples), >5 MeV ‘E’ (for B3, B4, and B5 glasses) up to 15 MeV ‘E’, all the obtained HVL, TVL, & MFP showed a decreasing trend. For instance, 2.279 cm, 7.569 cm, & 3.287 cm are the deduced respective HVL, TVL, & MFP values at 15 MeV ‘E’ for glass B5. As briefed for Figs. 4–7 results, here also, with ‘E’, all the noticed HVL, TVL, & MFP alterations are



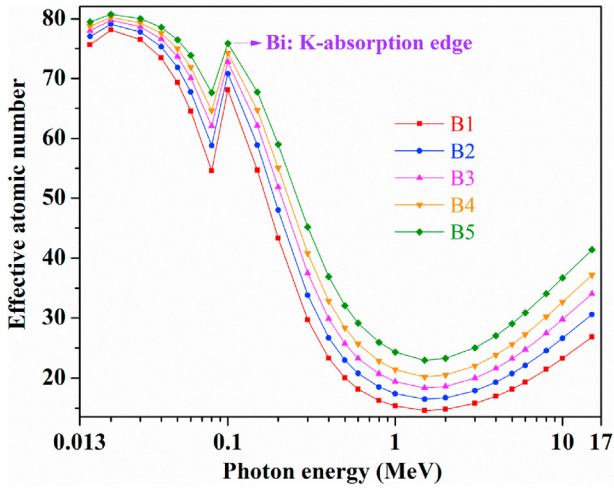


Fig. 6 – Variation of effective atomic number ( $Z_{eff}$ ) with photon energy (MeV) for all B1–B5 glasses.

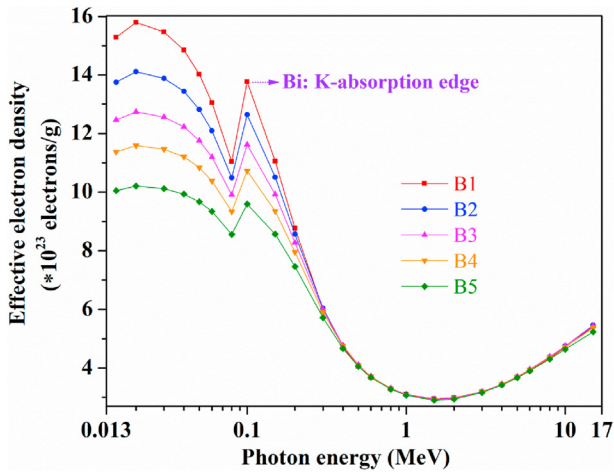


Fig. 7 – Variation of effective electron density ( $N_{eff}$ ) with photon energy (MeV) for all B1–B5 glasses.

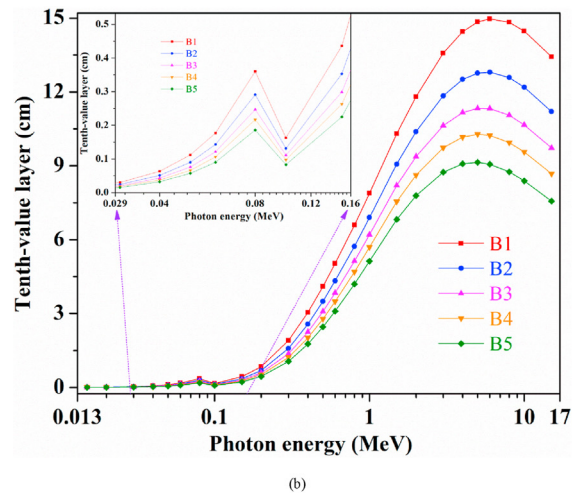
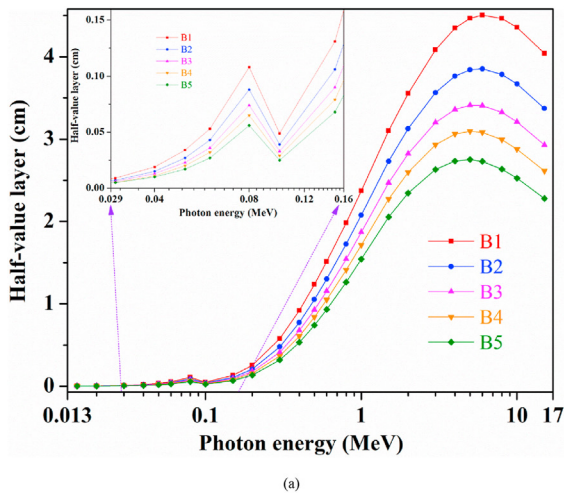
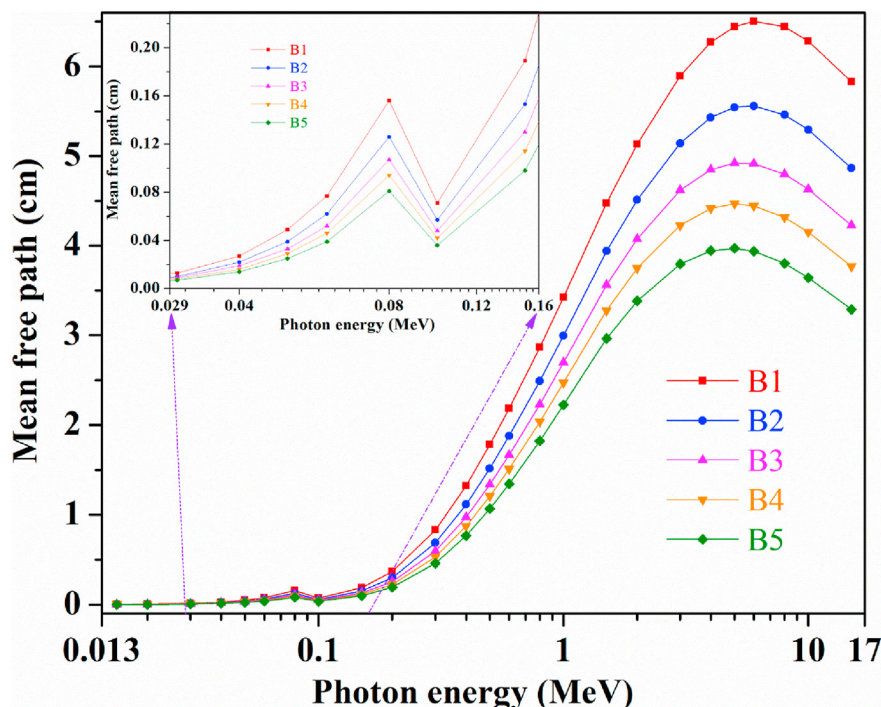


Fig. 8 – Variations of (a) half-value layer (HVL) and (b) tenth-value layer (TVL) with photon energy (MeV) (insets, within the 0.029–0.16 MeV photon energy range) for all B1–B5 glasses.

caused by PEA, CS, and PP actions preeminence at discrete ranges. Besides, seemingly, at all checked out energies, improving the  $\text{Bi}_2\text{O}_3$  amount from 20 → 42.5 mol% from B1→B5 glass has led to a decline in HVL, TVL, & MFP (see Figs. 8 and 9). Here, relatively less thickness is enough for glass B5 than remaining glasses to shield photons as ' $\rho$ ' enhances from B1→B5 glass (see Table 1). As an example, 0.053 cm, 0.043 cm, 0.036 cm, 0.032 cm, and 0.027 cm, & 0.077 cm, 0.062 cm, 0.052 cm, 0.046 cm, and 0.039 cm are the computed corresponding HVL & MFP for B1, B2, B3, B4, and B5 samples at 60 KeV 'E' which is commonly utilized in computed tomography (CT).

Relevantly, sample B5 HVL & MFP are related with reported respective quantities of commercial shielding glasses (SCHOTT AG: RS 253, RS 253 G18, RS 323 G19, RS 360, and RS 520) [44], SS403, CN, CS516, IL600, and MN400 alloys [45], natural rubber, polyacrylonitrile, polyethylacrylate, polyethylene tetraphthalate, polyoxymethylene, and polyphenyl methacrylate polymers [46], OC, BMC, HSC, IC, ILC, SMC, and SSC [47], and lead (Pb), including calcium silicide ( $\text{CaSi}_2$ ), magnesium silicide ( $\text{Mg}_2\text{Si}$ ), magnesium boride ( $\text{MgB}_2$ ), calcium hexaboride ( $\text{CaB}_6$ ), aluminum oxide ( $\text{Al}_2\text{O}_3$ ), and titanium dioxide ( $\text{TiO}_2$ ) ceramics [48], and are depicted in Fig. 10 (a-e) & Fig. 11 (a-e), accordingly. At 0.2 MeV, 0.662 MeV ( $^{137}\text{Cs}$ ), & 1.25 MeV ( $^{60}\text{Co}$ ) photon energies, sample B5 has lower HVL & MFP compared to all five commercial glasses values (see Fig. 10 (a) & Fig. 11 (a)). Besides, glass B5 owns evidently smaller HVL & MFP than compared all five forms of alloys up to 0.6 MeV 'E', except at 15 KeV 'E' (CN & MN400 alloys owns lesser values for 15 KeV 'E' than sample B5) concerning 0.015–15 MeV 'E' range (Fig. 10 (b) & Fig. 11 (b)). Also, within 'E' range of 15 KeV–15 MeV, in contrast to six kinds of polymers, seven types of concretes, and six forms of ceramics quantities, glass B5 possesses fewer HVL & MFP, respectively, except than Pb where Pb has minimal values at all tested energies (see Fig. 10 (c, d & e) & Fig. 11 (c, d, & e)). So, specifically, glass B5 with relatively less thickness is ample to effectively attenuate photons as opposed to distinct commercial glasses, polymers, concretes, and ceramics.

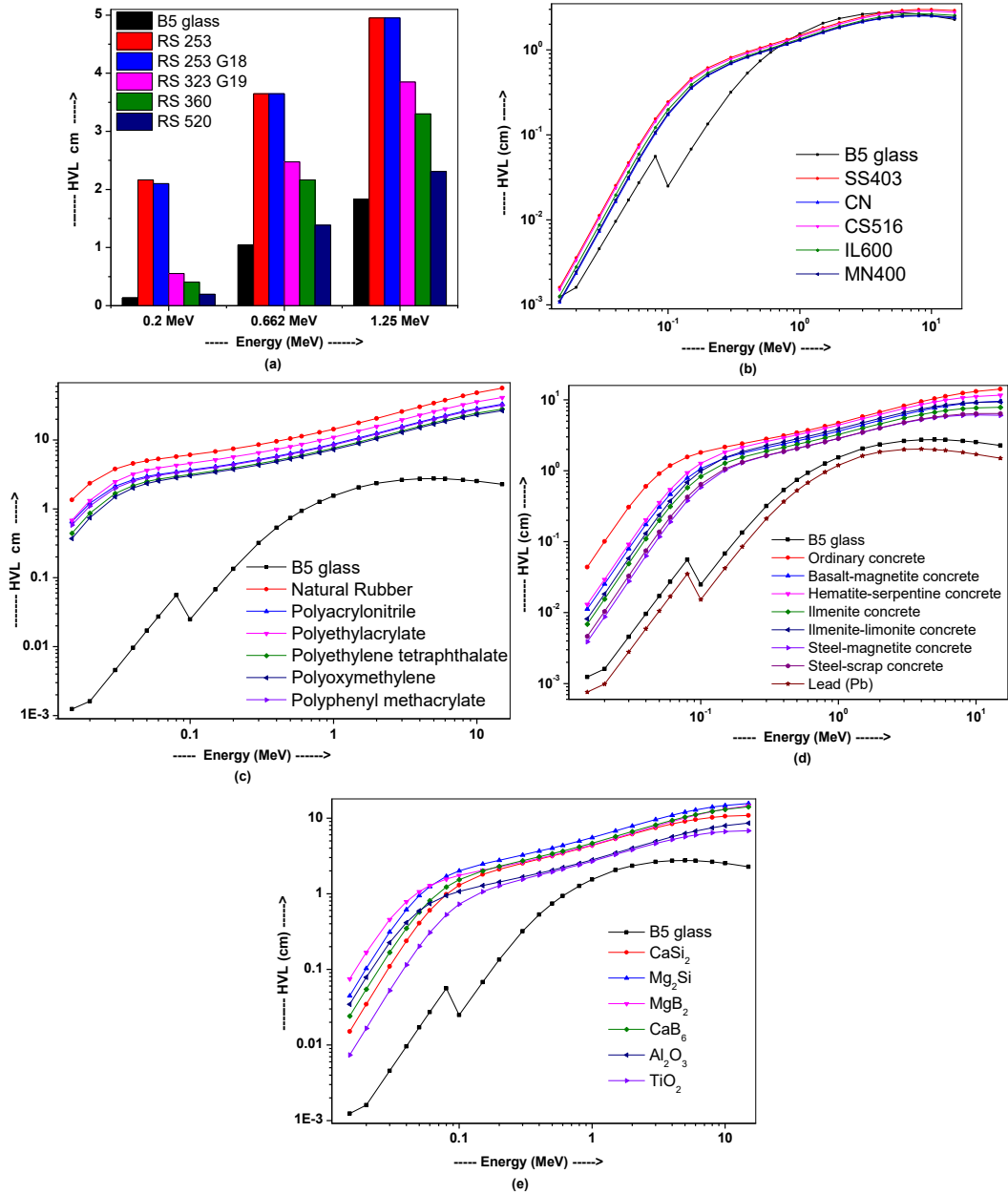


**Fig. 9** – Variation of mean free path (MFP) with photon energy (MeV) (inset, within the range of 0.029–0.16 MeV photon energy) for all B1–B5 glasses.

Corresponding Fig. 12 (a-e) & (f-j) present the reckoned EBFs & EABFs variations within inspected 'E' range at 1, 2, 5, 10, 15, 20, 25, 30, 35, and 40 mfp PDs for all B1–B5 glasses. For these five chosen samples, for calculated  $Z_{eq}$  & G–P fitting parameters (b, c, a,  $X_k$ , and d) data related to EBFs & EABFs computations within 0.015–15 MeV 'E' region readers' can refer to Table S1S5 in Supplementary Material. Following Fig. 12, one can observe three explicit sections of EBFs & EABFs as against 'E' which are principally relevant to PEA, CS, and PP phenomena, individually. Here, for all B1–B5 samples, EBFs & EABFs curves follow an identical movement with both 'E' & PD. Because of PEA, EBFs & EABFs for all B1–B5 glasses are minimal at lower 'E' range (0.015–0.3 MeV) as PEA eliminates photons from the glass, blocking them to buildup. But, at 0.02, 0.06, and 0.1 MeV 'E' i.e., near 'Bi' L1 ( $=0.0163875$  MeV) & K-absorption edges higher buildup of photons has transpired. Further, from B1→B5 sample, as 'Bi' improves from 56.1448 wt% → 74.6161 wt% (see Table 1), at 0.1 MeV 'E' peak, a gradually enhanced photons 'buildup' occurs (see Fig. 12). Later, > 0.3 up to 3 MeV 'E', for all studied glasses, EBFs & EABFs are regularly increased owing to the CS dominance as this mechanism causes to an only fractional reduction in 'E' by scattering and can not pluck out photons wholly, and here longer lifetimes of them make escape chances higher through the glass, resulting in larger EBFs & EABFs. Further, above 3 MeV up to 15 MeV 'E', the PP process is effective in all B1–B5 glasses, and due to this phenomenon EBFs & EABFs are enhanced at high energies and greater PDs (>10 mfp), except at smaller PDs like 1 & 2 mfp. However, the CS can not be ruled out completely and could

occur at all energies and in all substances. At bigger energies, secondary  $\gamma$ -rays are created due to complex scattering actions, leading to photons improved buildup. Also, with PD increment, consequently, thickness enlargement, commonly, in high- $Z_{eq}$  media these scatterings yields high  $\gamma$ -ray 'buildups' due to available large scattering volume. Glass B5 (high  $\text{Bi}_2\text{O}_3$  quantity, 42.5 mol%) by possessing greater  $Z_{eq}/Z_{eff}$  has the lower estimated EBFs & EABFs in all selected samples.

Within the tested 'E' range, for all B1–B5 samples ( $t = 7$  mm), the assessed RPE discrepancies are portrayed in Fig. 13. From this plot, one can observe that the  $\text{Bi}_2\text{O}_3$  content growth from 20 mol% (B1 sample) to 42.5 mol% (B5 glass) as against  $\text{B}_2\text{O}_3$  improves the photons attenuation feature, because as a high-Z element, Bi, incessantly aids the collisions between photons and glasses as opposed to 'B' (light element). Hence, in all selected glasses, sample B5 reveals analogously greater RPE within the studied 'E' region. The incident photons with 'E' 15–60 KeV can be totally soaked up in all B1, B2, B3, B4, and B5 glasses, even after, up to 0.15 MeV 'E' all these five samples have fairly large RPE, say, for 150 KeV 'E', exclusively, 97.52%, 98.97%, 99.54%, 99.78%, and 99.92%. Afterward, undeniably, for all B1–B5 samples, with advancing 'E', RPE reduced exponentially, proclaiming that the incoming larger energy photons are capable of pass over these glasses with ease. For instance, the RPE for B1 & B2 samples are 10.21% & 11.83%, respectively, for 6 MeV 'E' photons, and for 5 MeV 'E' penetrating  $\gamma$ -rays, for B3, B4, and B5 glasses, 13.26%, 14.51%, and 16.16% are the corresponding RPE. These results insinuate that, for example, glass B5 can absorb or attenuate at most

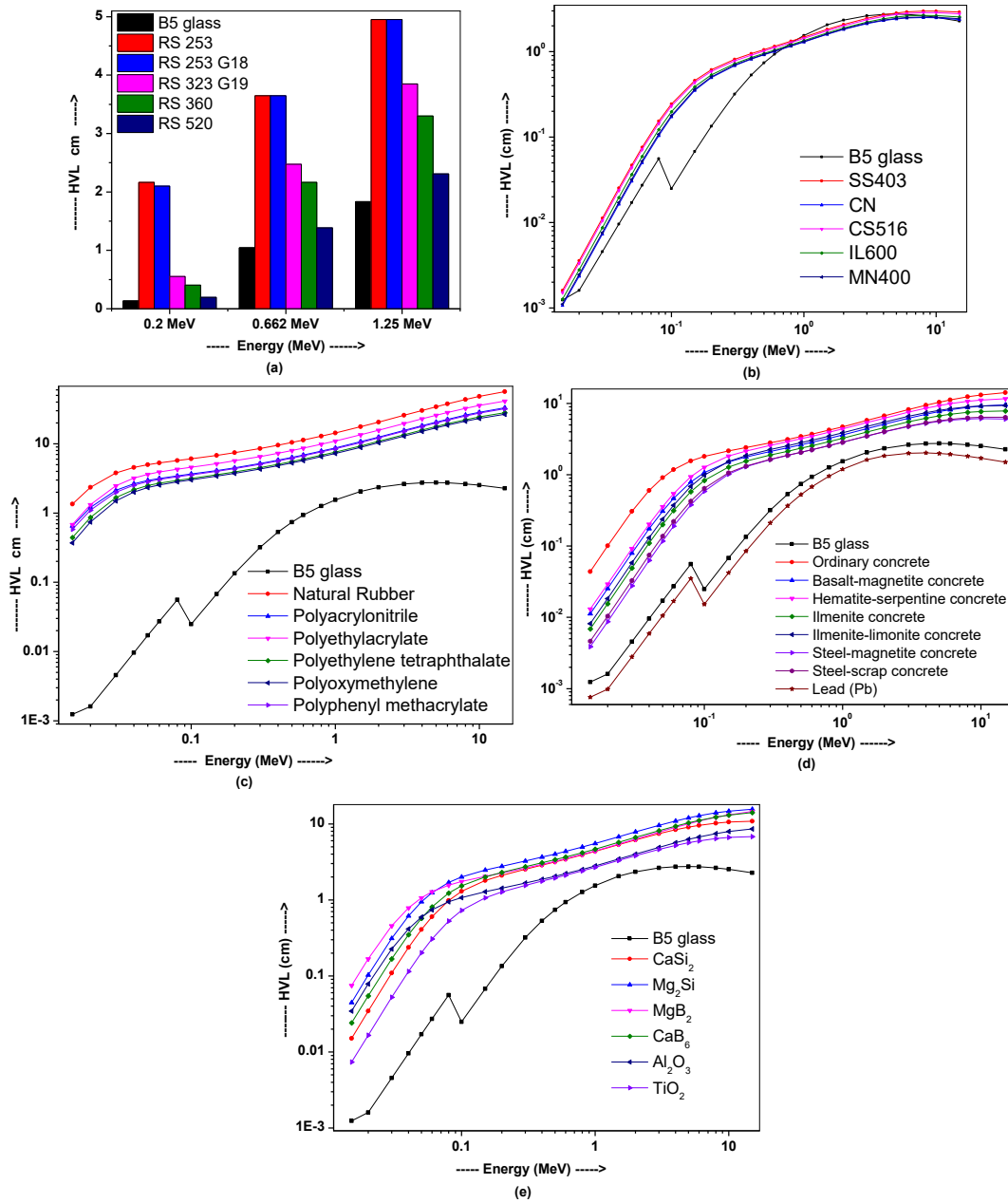


**Fig. 10 – Comparison of HVL of the glass ‘B5’ with some (a) commercial glasses, (b) alloys, (c) polymers, (d) standard shielding concretes and Lead, and (e) Ceramics.**

16.16% of the 5 MeV ‘E’ photons and the rest of 83.84% could go through it. Later, for 15 MeV ‘E’ photons, from B1 → B5 glass, 11.31%, 13.4%, 15.27%, 16.95%, and 19.18% are the discrete estimated RPE. From Fig. 13, conspicuously, the obtained RPE issues confirm selected  $B_2O_3$ – $Bi_2O_3$  glasses favorable photon shielding ability for lower energies in contrast to greater ones.

Fig. 14 (a) & (b) represents the corresponding estimated ‘ $\Psi_P$ ’ & ‘ $\Psi_A$ ’ divergences within the KE range of 15 KeV–15 MeV for all B1–B5 glasses. Here, one can observe that both  $\Psi_P$  (protons (charge = +1)) &  $\Psi_A$  ( $\alpha$ -particles (charge = +2)) patterns for all selected glasses show the same drifts with KE. As KE raises from 0.015 MeV both  $\Psi_P$  &  $\Psi_A$  are increased primarily. At

90 KeV for sample B1 & 100 KeV for glasses B2, B3, B4, and B5, maximal  $\Psi_P$  values are identified while at 0.7 MeV KE the highest  $\Psi_A$  quantities are noticed for all B1–B5 samples. Later, both  $\Psi_P$  &  $\Psi_A$  of all studied glasses are reduced consistently with additional KE increment up to considered 15 MeV. As  $\alpha$ -particles possess larger mass ( $=6.645 \times 10^{-24}$  g), consequently, smaller speed relative to protons (mass =  $1.673 \times 10^{-24}$  g), greater MSPs are perceived for  $\alpha$ -particles and also these particles reach the most at bigger KE compared to protons as  $\alpha$ -particles lose less KE (electronic excitation & ionization of atoms) by interacting with outer orbital electrons of atoms in glasses. Here, sample B5, which owns the highest ‘Bi’ element



**Fig. 11 – Comparison of MFP of the glass ‘B5’ with some (a) commercial glasses, (b) alloys, (c) polymers, (d) standard shielding concretes and Lead, and (e) Ceramics.**

weight, 74.6161 wt%, exhibits relatively lower  $\Psi_p$  &  $\Psi_A$  in all B1–B5 glasses, i.e., more competent in ceasing bombarding protons and  $\alpha$ -particles, which can also be connected to its heavier ‘ $\rho$ ’ (high  $\text{Bi}_2\text{O}_3$  add-on).

Likewise, for all B1–B5 glasses, Fig. 15 (a) & (b) shows calculated  $\Phi_p$  &  $\Phi_A$  changes graphically within 0.015–15 MeV KE region, accordingly, and the respective inset plots display the zoom-in KE range at –0.02–2.1 MeV. Generally, for best proton &  $\alpha$ -particle freezing potential, lower ‘PRs’ are imperative for materials. It is apparent from Fig. 15 results that with KE, both the  $\Phi_p$  &  $\Phi_A$  gradationally reduced as  $\text{Bi}_2\text{O}_3$  content increases from sample B1→B5 in reference to  $\text{B}_2\text{O}_3$

implying that sample B5 possesses the better terminating ability for protons &  $\alpha$ -particles. Besides, from 15 KeV up to 15 MeV KE both  $\Phi_p$  &  $\Phi_A$  values of all studied glasses have grown nearly linearly. Here, it is identified that the range of  $\alpha$ -particles is shorter in contrast to that of protons and basically  $\sim 1/9$  quantity of  $\Phi_p$  is reasonable to attenuate the incoming  $\alpha$ -particles. At the explored KE range, for glass B5, the computed  $\Phi_p$  &  $\Phi_A$  quantities are varied within 0.1381–992.42  $\mu\text{m}$  & 0.0935–112.12  $\mu\text{m}$  ranges, respectively.

Accessorially, within the KE range of 15 KeV–15 MeV, for all B1–B5 samples, corresponding Fig. 16 (a) & (b) illustrates the derived ‘ $\Psi_E$ ’ & CSDA range variabilities whereas the

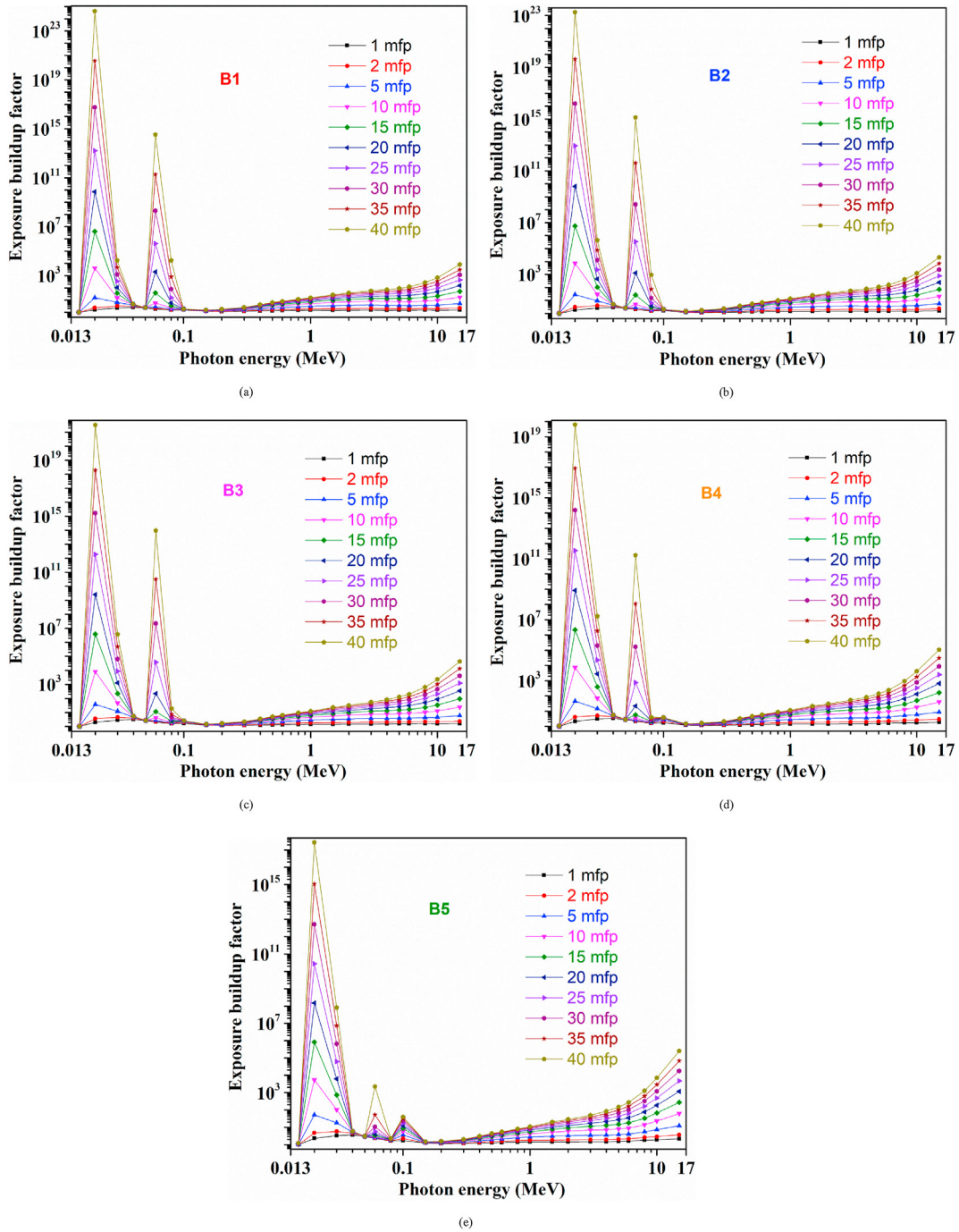


Fig. 12 – Variations of (a–e) exposure buildup factor (EBF) and (f–j) energy absorption buildup factor (EABF) with photon energy at different mean free paths for all B1–B5 glasses.

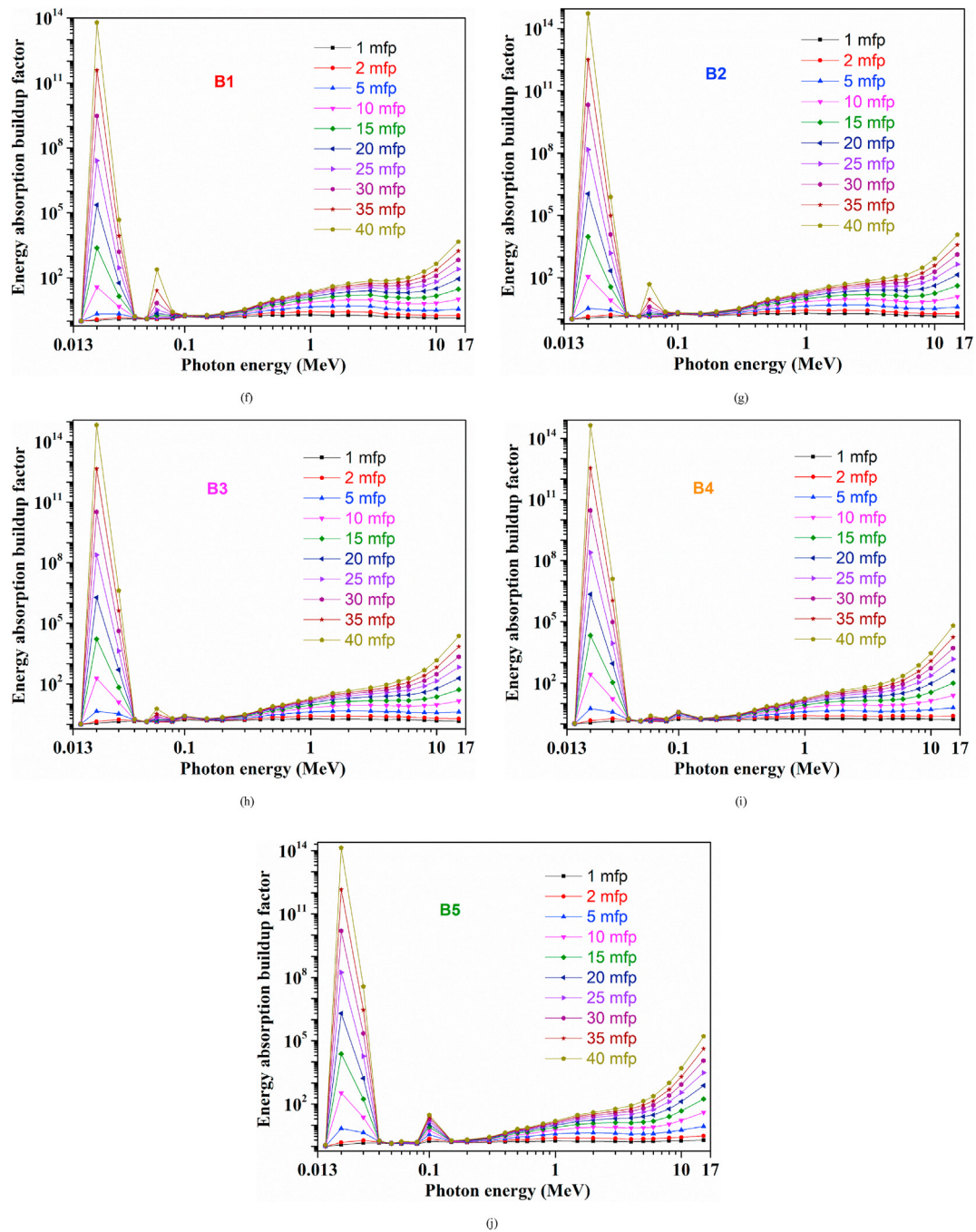


Fig. 12 – (continued).

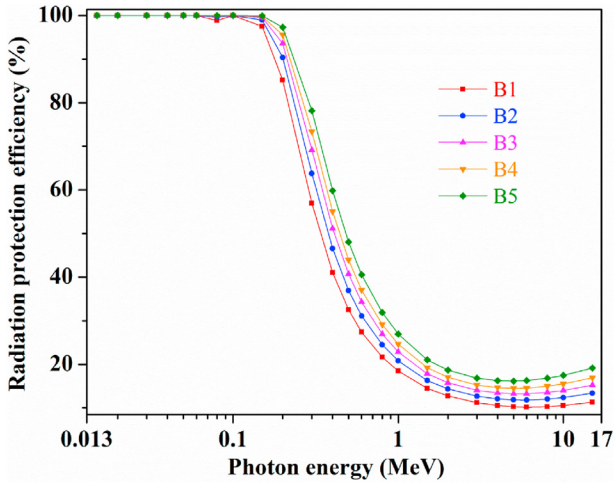


Fig. 13 – Variation of radiation protection efficiency (RPE) with photon energy (MeV) for all B1–B5 samples.

respective inset plots exhibit expanded KE ranges at 0.9–15.5 MeV & 0.014–0.21 MeV. Following Fig. 16 (a) one can notice that with the electron KE, ‘ $\Psi_E$ ’ decline from sample B1→B5, and has the lowest values at 0.9 MeV for B1– B4 glasses, and 0.8 MeV for sample B5 with enhancing KE. Then, ‘ $\Psi_E$ ’ quantities increase nominally up to 15 MeV KE for all B1–B5 glasses.

However, the CSDA ranges increase with electron KE though the ‘ $\Psi_E$ ’ reduces (see Fig. 16 (b)). Usually, at lower KE, the electrons very frequently collide with low-Z compounds owing to high  $N_{eff}$  (see Fig. 7) while with enhancing KE more interactions of electrons may happen for high-Z substances [27]. Here, B1 glass is influential to shield low KE electrons whereas sample B5 is effective in attenuating greater KE electrons which is on grounds of the Bremsstrahlung radiation.

As a consequence of regulated nuclear fission reactions (heavier nucleus progression into lighter nuclei), commonly, neutrons are created. Banking on KE, neutrons are

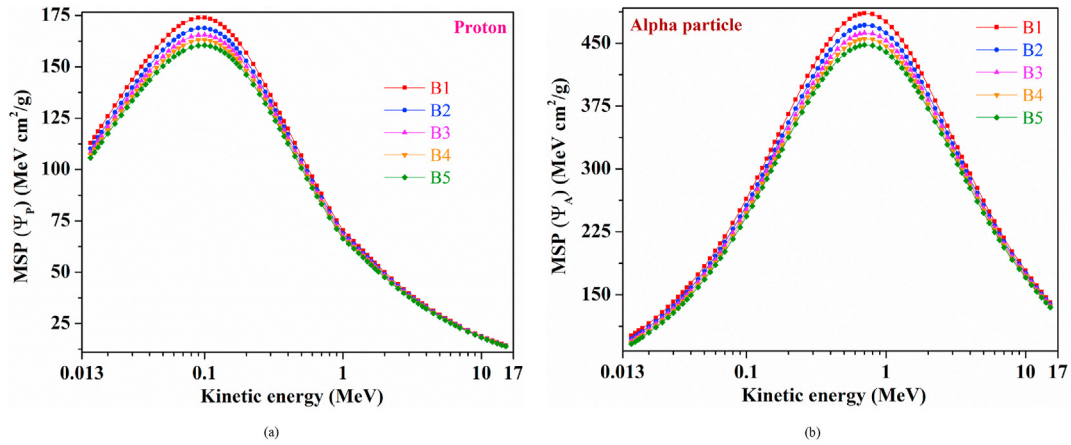


Fig. 14 – Variations of (a) proton mass stopping power ( $\Psi_P$ ) and (b) alpha mass stopping power ( $\Psi_A$ ) as a function of kinetic energy (KE) for all B1–B5 glasses.

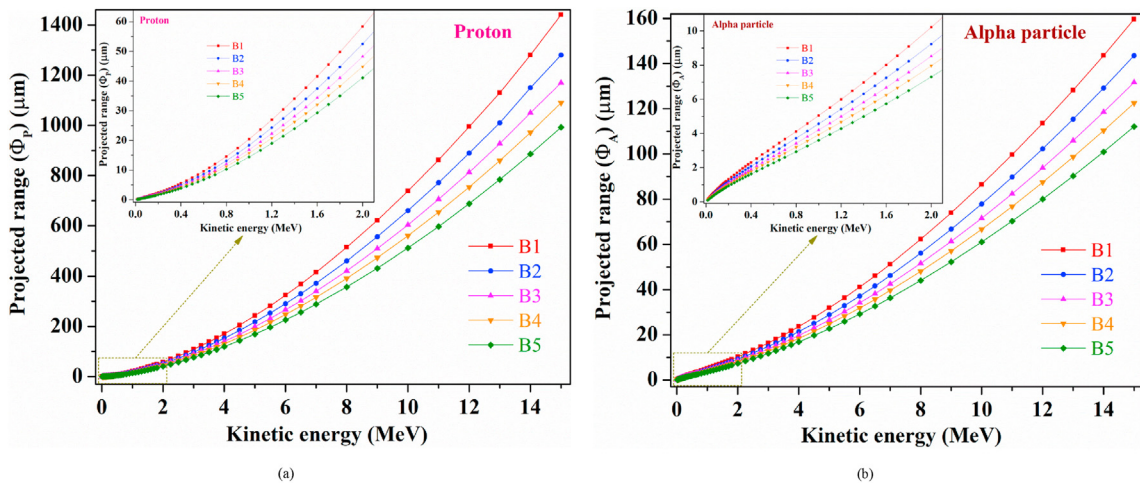


Fig. 15 – Variations of (a) proton projected range ( $\Phi_P$ ) and (b) alpha projected range ( $\Phi_A$ ) as a function of kinetic energy (KE) (insets, within KE regions at 0.02–2.1 MeV) for all B1–B5 glasses.

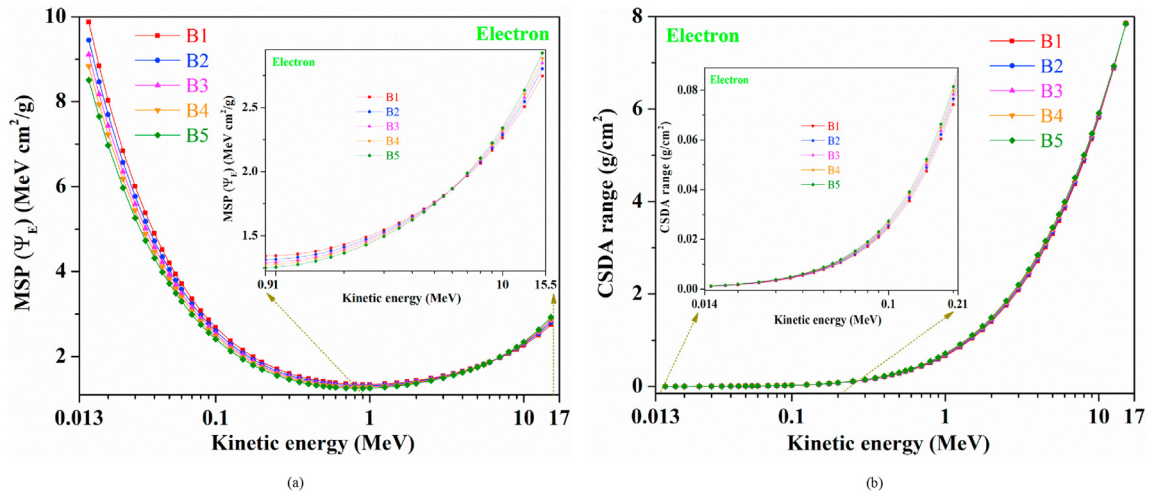


Fig. 16 – Variations of (a) electron stopping power ( $\Psi_E$ ) (inset, within the range of 0.9–15.5 MeV KE) and (b) CSDA range (inset, within the 0.014–0.21 MeV KE range) as a function of kinetic energy (KE) for all B1–B5 samples.

Table 3 – Effective removal cross-sections for fast neutrons  $\Sigma_R$  ( $\text{cm}^{-1}$ ) for all B1–B5 glasses.

Glass Code	Element	$\Sigma_R/\rho$ ( $\text{cm}^2/\text{g}$ )	Fraction by weight (%)	Partial Density ( $\text{g}/\text{cm}^3$ )	$\Sigma_R$ ( $\text{cm}^{-1}$ )
B1	B	0.0575	0.116179	0.499802058	0.02873861834
	O	0.0405	0.322373	1.386848646	0.05616737016
	Bi	0.0103	0.561448	2.415349296	0.02487809775
	$\Sigma_R = 0.1097840863$				
Total $\Sigma_R$ for glass 'B1' = 0.10978 $\text{cm}^{-1}$					
B2	B	0.0575	0.096124	0.468412252	0.02693370449
	O	0.0405	0.284505	1.386392865	0.05614891103
	Bi	0.0103	0.619371	3.018194883	0.03108740729
	$\Sigma_R = 0.1141700228$				
Total $\Sigma_R$ for glass 'B2' = 0.11417 $\text{cm}^{-1}$					
B3	B	0.0575	0.080285	0.431853015	0.02483154836
	O	0.0405	0.254598	1.369482642	0.055464047
	Bi	0.0103	0.665117	3.577664343	0.03684994273
	$\Sigma_R = 0.1171455381$				
Total $\Sigma_R$ for glass 'B3' = 0.11714 $\text{cm}^{-1}$					
B4	B	0.0575	0.067459	0.393555806	0.02262945885
	O	0.0405	0.230381	1.344042754	0.05443373154
	Bi	0.0103	0.70216	4.09640144	0.04219293483
	$\Sigma_R = 0.1192561252$				
Total $\Sigma_R$ for glass 'B4' = 0.11926 $\text{cm}^{-1}$					
B5	B	0.0575	0.052224	0.336374784	0.01934155008
	O	0.0405	0.201615	1.298602215	0.05259338971
	Bi	0.0103	0.746161	4.806023001	0.04950203691
	$\Sigma_R = 0.1214369767$				
Total $\Sigma_R$ for glass 'B5' = 0.12144 $\text{cm}^{-1}$					

categorized as thermal (0.0253 eV KE) or slow neutrons (<0.5 eV KE), epithermal or medium neutrons (up to 5 KeV KE), and fast neutrons (>500 KeV KE) [5]. Table 3 presents the ' $\Sigma_R$ ' calculation steps and the correlated  $\Sigma_R$  quantities for all B1–B5 glasses. 0.10978  $\text{cm}^{-1}$ , 0.11417  $\text{cm}^{-1}$ , 0.11714  $\text{cm}^{-1}$ , 0.11926  $\text{cm}^{-1}$ , and 0.12144  $\text{cm}^{-1}$ , respectively, are the obtained  $\Sigma_R$  for B1, B2, B3, B4, and B5 samples. Here, glass B5 (B: 5.2224 wt%, Bi: 74.6161 wt%, O: 20.1615 wt%) possesses the largest  $\Sigma_R$  in all selected glasses, signifying that it has correspondingly greater efficacy to shield fast neutrons. This

Table 4 – Comparison of  $\Sigma_R$  ( $\text{cm}^{-1}$ ) of glass B5 with reported various nuclear radiation shielding materials.

Sample	$\Sigma_R$	Reference
'B5' glass	0.12144	Present work
Graphite (C)	0.0773	[49]
Water ( $\text{H}_2\text{O}$ )	0.1023	
Polyethylene grains	0.0789	
Ethanol ( $\text{C}_2\text{H}_6\text{O}$ )	0.0939	
$\text{B}_4\text{C}$	0.0714	
Ordinary concrete (OC)	0.0937	[47]
Hematite-serpentine concrete (HSC)	0.0967	
Ilmenite-limonite concrete (ILC)	0.0950	
Basalt-magnetite concrete (BMC)	0.1102	
Ilmenite concrete (IC)	0.1121	
Steel-scrap concrete (SSC)	0.1247	
Steel-magnetite concrete (SMC)	0.1420	
TZN-D glass	0.108	[25]
SCNZ5 glass	0.053077264	[26]
H5 glass	0.138	[27]
TBBT30 glass	0.116862332	[28]
L15 glass	0.104	[29]
Fe-05 composite	0.0924	[50]
Pd50Mn50 alloy	0.161	[51]
Polyethersulfone (PES) polymer	0.088	[52]

means, though  $^{10}\text{B}$  atom owns a large neutron absorption cross-section ( $\sigma_A = 760 \text{ b}$ ,  $1 \text{ b} = 10^{-28} \text{ m}^2$ ), heavy element 'Bi' also takes part in boosting  $\Sigma_R$  (see Table 3) as for any certain medium, high ' $\rho$ ' (e.g., sample B5,  $\rho = 6.441 \text{ g}/\text{cm}^3$ ) is desirable in fast neutrons shielding, requiring both lighter & heavier elements appropriate mixtures in it. Glass B1 holds the minimum  $\Sigma_R$  relying on the according B, Bi, and O elements whole participations in it.

Supplementarily, glass B5 ' $\Sigma_R$ ' is correlated with corresponding alternative classical neutron shields and some other just recently reported distinct radiation shielding materials  $\Sigma_R$  [25–29, 47, 49, 50–52] and the related outcomes are presented in Table 4. Here, sample B5 has greater  $\Sigma_R$  than C,  $\text{H}_2\text{O}$ , Polyethylene grains,  $\text{C}_2\text{H}_6\text{O}$ , &  $\text{B}_4\text{C}$  [49], OC, HSC, ILC, BMC, & IC [47], and TZN-D [25], SCNZ5 [26], TBBT30 [28], & L15 [29] glasses, Fe-



05 composite [50], and PES polymer [52], and a lower quantity in contrast to SSC & SMC [47], H5 glass [27], and Pd50Mn50 alloy [51]  $\Sigma_R$ . In our previous work, for GeO<sub>2</sub>-Tl<sub>2</sub>O-Bi<sub>2</sub>O<sub>3</sub> glass system, appropriate  $\gamma$  & neutron shielding aspects were reported [53].

#### 4. Conclusions

In the current study, directly and indirectly ionizing radiations ( $\gamma$ , proton,  $\alpha^-$ ,  $e^-$ , & neutron) attenuation competences for chosen five bismuth borate glasses were explored by scrutinizing all appropriate  $\mu$ ,  $\mu/\rho$ ,  $Z_{eff}$ ,  $N_{eff}$ , HVL, TVL, MFP,  $Z_{eq}$ , EBF, EABF, RPE,  $\Psi_P$ ,  $\Phi_P$ ,  $\Psi_A$ ,  $\Phi_A$ ,  $\Psi_E$ , CSDA ranges, and  $\Sigma_R$  variables. Theoretically (by Phy-X/PSD) deduced  $\mu/\rho$  quantities were validated with computationally (utilizing respective MCNPX, Geant4, and FLUKA codes) obtained  $\mu/\rho$  outputs for all B1–B5 glasses. For instance, at 15 KeV energy, 65.3893, 72.1961, 77.1494, 81.7293, and 86.298 cm<sup>2</sup>/g were  $\mu/\rho$  (using FLUKA code), correspondingly, for B1, B2, B3, B4, and B5 samples while for the same glasses at the same energy, 65.753, 72.391, 77.633, 81.878, and 86.92 cm<sup>2</sup>/g were the respective calculated  $\mu/\rho$  values applying Phy-X/PSD. Bi<sub>2</sub>O<sub>3</sub> insertion in place of B<sub>2</sub>O<sub>3</sub> demonstrated an advantageous impact on  $\gamma$ -ray shielding abilities (enhanced  $Z_{eff}$  and a reduction in HVL, TVL, & MFP) in all B1–B5 glasses, at any selected photon energy. For example, at 15 MeV energy, for glass B5, which owns the lowest HVL, TVL, & MFP in all B1–B5 samples, 2.279 cm, 7.569 cm, and 3.287 cm were the derived according values. Also, proportionately, sample B5 has smaller HVL & MFP than corresponding quantities of five commercial glasses at 0.2, 0.662, and 1.25 MeV  $\gamma$ -ray energies, and six types of polymers, seven kinds of concretes & six forms of ceramics within photon energy range of 0.015–15 MeV. Moreover, at 1–40 mfp with selected 10 PDs, for assessed EBFs & EABFs, for all B1–B5 glasses within 15 KeV–15 MeV photon energy range, at lower, medium, and larger energy ranges individual PEA, CS, and PP mechanisms supremacy was recognized. Additionally, for all selected samples,  $\Psi_P$ ,  $\Phi_P$ ,  $\Psi_A$ ,  $\Phi_A$ ,  $\Psi_E$ , and CSDA range for electrons values within 0.015–15 MeV KE range were estimated. Further, in all B1–B5 glasses, the B5 sample (Bi: 74.6161 wt% & B: 5.2224 wt%) exhibits relatively higher  $\Sigma_R$  (=0.12144 cm<sup>-1</sup>) as Bi<sub>2</sub>O<sub>3</sub> improves the glasses ' $\rho$ ' (4.302 g/cm<sup>3</sup> → 6.441 g/cm<sup>3</sup>). Comprehensively, larger ' $\rho$ ' (=6.441 g/cm<sup>3</sup>), bigger  $\mu$ ,  $\mu/\rho$ ,  $Z_{eff}$ ,  $Z_{eq}$ , RPE, & minimal HVL, TVL, MFP, EBFs, EABFs, fewer ranges for protons,  $\alpha^-$ , and  $e^-$ , greater  $\Sigma_R$  of glass B5 among all inspected glasses revealed its better shielding capacity for all  $\gamma$ , proton,  $\alpha^-$ , and  $e^-$  radiations, and fast neutrons. Thus, glass B5 can be utilized as a nuclear waste storage container and as a potential radiation shielding medium at nuclear reactors and nuclear medicine facilities.

#### CRedit author statement

G. Lakshminarayana: Conceptualization, Visualization, Writing - Original Draft, Writing - Review and Editing, Supervision Ashok Kumar: Software, Formal analysis, Data Curation, Validation H.O. Tekin: Software, Formal analysis, Data Curation Shams A.M. Issa: Software, Formal analysis, Data

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#### Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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#### Appendix A. Supplementary data

Supplementary data to this article can be found online at <https://doi.org/10.1016/j.jmrt.2020.10.019>.

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